Diffusive dynamics during the high-to-low density transition in amorphous ice


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Water exists in high- and low-density amorphous ice forms (HDA and LDA), which could correspond to the glassy states of high-(HDL) and low-density liquid (LDL) in the metastable part of the phase diagram. However, the nature of both the glass transition and the high-to-low-density transition are debated and new experimental evidence is needed. Here we combine wide-angle X-ray scattering (WAXS) with X-ray photon-correlation spectroscopy (XPCS) in the small-angle X-ray scattering (SAXS) geometry to probe both the structural and dynamical properties during the high-to-low-density transition in amorphous ice at 1 bar. By analyzing the structure factor and the radial distribution function, the coexistence of two structurally distinct domains is observed at $T = 125$ K. XPCS probes the dynamics in momentum space, which in the SAXS geometry reflects structural relaxation on the nanometer length scale. The dynamics of HDA are characterized by a slow component with a large time constant, arising from viscouselastic relaxation and stress release from nanometer-sized heterogeneities. Above 110 K a faster, strongly temperature-dependent component appears, with momentum transfer dependence pointing toward nanoscale diffusion. This dynamical component slows down after transition into the low-density form at 130 K, but remains diffusive. The diffusive character of both the high- and low-density forms is discussed among different interpretations and the results are most consistent with the hypothesis of a liquid-liquid transition in the ultraviscous regime.

Significance

The importance of a molecular-level understanding of the properties, structure, and dynamics of liquid water is recognized in many scientific fields. It has been debated whether the observed high- and low-density amorphous ice forms are related to two distinct liquid forms. Here, we study experimentally the structure and dynamics of high-density amorphous ice as it relaxes into the low-density form. The unique aspect of this work is the combination of two X-ray methods, where wide-angle X-ray scattering provides the evidence for the structure at the atomic level and X-ray photon-correlation spectroscopy provides insight about the motion at the nanoscale, respectively. The observed motion appears diffusive, indicating liquid-like dynamics during the relaxation from the high-to-low-density form.


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being the momentum transfer. In the SAXS geometry, we use

\[ Q \approx 1.7 \text{ Å}^{-1} \]

and second coordination shell (\( r \approx 3.5 \text{ Å} \)), indicating loss of interstitial molecules and increased tetrahedral contribution in the low-density form (33) with an isosbestic point at 4.15 Å.

The complex shift observed in uHDA is attributed to strain release and large interface scattering of interpenetrating structures, and the transition to the low-density form involves several intermediate states (32). In eHDA the observed double peak in \( S(Q) \) indicates coexistence of high- and low-density domains, as suggested by earlier experimental (5, 7) and theoretical investigations (15). In this case, the decomposition of both the static structure factor and the radial distribution function into the sum of two components can be justified by assuming that the contribution of interfaces to the scattering is negligible due to the large domain sizes. The temperature dependence of \( S(Q) \) (Fig. 2) signifies the transition of HDA to a low-density form. The question is whether this transition occurs between amorphous ice forms, i.e., an HDA-to-LDA transition, or whether the transition occurs between two liquid states, inferring instead an HDL-to-LDL transition in the ultrasonic regime. Another possibility would be that the transition involves both the liquid and glassy forms (such as HDL→LDA→HDA or HDL→LDA→LDL). Such cases cannot be distinguished purely from the static structure factor and therefore dynamical information like the intermediate scattering function accessible by XPCS is needed.

**Dynamical Information from XPCS**

The XPCS data at SAXS geometry were measured at a photon energy of 8.4 keV (Materials and Methods). The pronounced signal at small scattering angles of \( Q \approx 0.01 \text{ Å}^{-1} \) reflects nanometer length-scale heterogeneities presumably due to the grainy nanostructure of the sample. Previous neutron-scattering studies discuss structural heterogeneities as a result of scattering in the Porod-limit region (13) (\( Q \sim 0.1 \text{ Å}^{-1} \)), which is beyond the scope of the extended static structure factor \( S(0) \) and radial distribution function \( g(r) \). (A) Comparison of the structure factor of the uHDA and annealed eHDA forms. The black solid line highlights the difference in behavior between eHDA with an isosbestic point and uHDA with a continuously shifting peak. (B) The extended static structure factor \( S(0) \) at different temperatures, during the transition from the high- to low-density form. (C) The amplitude of \( S(0) \) at \( Q = 1.7 \text{ Å}^{-1} \) (red circles) and \( Q = 2.1 \text{ Å}^{-1} \) (blue squares) as a function of temperature. (D) The radial distribution functions \( g(r) \) for different temperatures, where the isosbestic point at \( r = 4.15 \text{ Å} \) is indicated by the arrow.

![Fig. 1. Coherent X-ray diffraction setup combining WAXS with XPCS in SAXS. (A) The experimental scattering geometry used employs two detectors for WAXS and SAXS, respectively. (B) The WAXS scattering pattern allows distinguishing high- and low-density forms; the lines indicate the position of the maxima at momentum transfer \( Q = 1.7 \text{ Å}^{-1} \) and 2.1 Å\(^{-1} \). (C) The scattering pattern recorded in SAXS geometry is used to extract the dynamics with XPCS. Here the speckle patterns for different exposure times at temperature \( T = 89 \text{ K} \) are shown.](image-url)
of the present study. The present XPCS measurement is performed in the \( Q \) range of 0.003–0.01 Å\(^{-1}\), a wave-vector range inaccessible to visible light (34), which is connected to structural rearrangements on the \( \sim 100\)-nm length scale (35, 36). The dynamics are recorded by measuring the temporal autocorrelation function \( g_2(Q, \delta t) \) where \( \delta t \) is the correlation time, which can be related to the intermediate scattering function \( F(Q, \delta t) \) through

\[
g_2(Q, \delta t) = \beta |F(Q, \delta t)|^2 + 1,
\]

where \( \beta \) is the speckle contrast that depends on the experimental parameters.

Fig. 3 A and B shows the WAXS scattering patterns that were recorded giving the static structural information and the \( |F(Q, \delta t)|^2 \) dynamics measured in SAXS geometry. Here, the displayed \( I(Q) \) and \( |F(Q, \delta t)|^2 \) shown were recorded at the same sample position, using two detectors at different scattering angles.

The \( |F(Q, \delta t)|^2 \) data are fitted with a double-exponential function (Materials and Methods), which at \( T = 89 \) K yields a single time constant, whereas at temperatures \( T > 110 \) K it captures an additional faster component. The WAXS scattering pattern at 130 K changed during the XPCS measurement (Fig. 3A), as indicated by the pattern recorded before (solid line) and after (dashed line) the XPCS measurement. In Fig. 3B, at 130 K the solid and dashed lines depict the fits at early and late times, respectively, during the high-to-low-density transition (the first 0–400 s and last 700–1,000 s of the total 1,000 s recorded), which is accompanied by a heat release during the transformation (SI Appendix).

Two-time temporal correlation maps \( C(Q, \delta t, t) \) (28, 37–40) (Fig. 3C; Materials and Methods) provide a very sensitive measure of the underlying dynamics and allow resolving whether the system exhibits dynamical heterogeneities, or if the recorded dynamics are homogeneous. In the former case the linewidth of \( C(Q, \delta t, t) \) would fluctuate during the measurement time \( t \); in the latter, the width would remain constant. For \( T = 89 \) K the \( C(Q, \delta t, t) \) exhibits fluctuations with a multitude of relaxation times, whereas at \( T = 125 \) K, the dynamics speed up during the measurement. At 130 K dynamics are initially fast and more homogeneous (region 1) but after ~400 s a transition to slower and more heterogeneous dynamics is observed (regions 2 and 3).

The time constants \( \tau \) obtained by the double-exponential fit are shown in Fig. 4A. The slower component appears nearly temperature-independent, whereas the faster component manifested at \( T = 115 \) K accelerates ~100-fold upon heating to \( T = 130 \) K. The relative amplitudes of the two components change with increasing temperature \( T > 110 \) K, with the slow component gradually decaying and the fast component gaining in amplitude, shown in SI Appendix.

The solid and dashed lines in Fig. 4A are obtained from Arrhenius fits (Materials and Methods). A similar trend is seen by the normalized variance \( \chi^2 \) extracted from the \( C(Q, \delta t, t) \) (38), which is an experimental estimator of the four-point dynamical susceptibility (41). The amplitude of \( \chi^2 \) of \( C(Q, \delta t, t) \) (Fig. 4B) is proportional to the relative width \( \Delta \tau / \tau \) of the characteristic relaxation time; the maximum peak value \( \chi_{20} \) is thus a measure of the dynamical heterogeneities in the system. Fig. 4B (Inset) shows a abrupt decay in \( \chi_{20} \) for \( T > 115 \) K, indicating a transition to a more dynamically homogeneous regime.

With XPCS in the low-\( Q \) regime one can derive the diffusion constant on nanometer length scales by the \( Q \) dependence of the characteristic time constant (29). For hyperdiffusion or ballistic

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**Fig. 3.** Dynamical information obtained through the intermediate scattering function \( F(Q, \delta t) \). (A) The angularly integrated intensity in WAXS geometry at different temperatures. The solid and dashed lines at \( T = 130 \) K indicate the scattering intensity recorded in the beginning and at the end of the measured dynamics, respectively. (B) The intermediate scattering function \( |F(Q, \delta t)|^2 \) at different temperatures recorded in SAXS geometry at \( Q = 0.01 \) Å\(^{-1}\). The solid and dashed lines at \( T = 130 \) K indicate double-exponential fits recorded during the initial 400 s (solid line) and final 300 s (dashed line) of the total 1,000-s trace. (C) The two-time temporal autocorrelation \( C(Q, \delta t, t) \) for different temperatures at \( Q = 0.01 \) Å\(^{-1}\). At lower temperatures, a large variation of timescales is observed along \( \delta t \). At \( T = 125 \) K one can see a gradual acceleration indicated by the dashed line. At \( T = 130 \) K the dynamics change during the measurement and the dashed lines indicate regions (1), (2), and (3), which are analyzed separately.
motion $\tau$ exhibits $1/\tau \sim Q$ dependence, whereas $1/\tau \sim Q^2$ dependence is typical for diffusion, with the diffusion constant $D_0$ as the proportionality constant. Fig. 4C shows the $1/\tau$ versus $Q^2$ for temperatures of 115 K and above. At temperatures below 115 K and for the slow component no $Q^2$ dependence is observed. This observation indicates that the slow component is not diffusive and can be associated with viscoelastic relaxation due to release of inner stresses and convective flow (29, 42). From the fits in Fig. 4C $D_0$ is determined for the fast component to vary from 0.8 to 47.5 nm$^2$/s at 115–130 K. No corresponding faster fluctuation or diffusive dynamics are observed in crystalline ice as discussed in SI Appendix, Fig. S8).

Discussion

It is essential to discuss whether the experimental evidence provides further insight about the glass transition and whether the observed high-to-low-density transition occurs between amorphous or liquid states. The glass-transition temperature $T_g$ depends on the heating rate (SI Appendix). Previous investigations using calorimetry proposed $T_g\text{HDA} \sim 116$ K (9) using a heating rate of 10 K/min. In dielectric spectroscopy measurements, where a much slower heating rate was used, the glass-transition temperature for HDA was determined at $T_g\text{HDA} = 110$ K; the present heating rate is comparable to that used for the dielectric spectroscopy measurements. Therefore, we propose that the observed faster component at temperatures above 110 K (Fig. 4A), accompanied by the decrease of $\chi_0$, indicate a glass-to-liquid transition from HDA to HDL. Because the observed glass transition is kinetic in nature, the diffusivity exhibits continuous behavior during the transition. Therefore, we conclude that the reason for not observing the diffusive component below 110 K is that it occurs on a timescale longer than the experimental observation time, where the dynamics are dominated by the viscoelastic component. The acceleration of the dynamics at 125 K (Fig. 3C) is due to the ongoing HDA–HDL transition within the 1,000-s duration of the XPCS measurement. Because XPCS is measured in the SAXS geometry, it is sensitive to diffusion occurring due to structural rearrangements of the nanoscale heterogeneities. The diffusion involves translational motion at the nanometer length scale and therefore we can determine that the fast component does not reflect purely molecular reorientation, responsible for an orientational glass transition (25). However, there can be reorientation occurring, coupled to translational motion (43). We also conclude that the observed diffusive behavior near the glass transition would not manifest itself in a crystal-like amorphous ice (14, 26), but instead indicates that the amorphous matrix exhibits HDL-like dynamics.

In Fig. 4D, the dynamical properties are analyzed in three different time domains during the high-to-low-density transition. This transition is kinetically driven and accelerated due to the heat release, which is associated with a temperature jump of the order of 10 K (SI Appendix) and will therefore lead to a step-like change between regions 1 and 2. Directly after the jump between regions 1 and 2 (Fig. 4D), both $\tau$ and $D_0$ increase and $D_0$ decreases, followed by a more significant change in region 3, whereas a similar behavior is seen from the angularly integrated SAXS signal (SI Appendix, Fig. S6). The data indicate that the transformation to the low-density form is completed upon entering region 2. The corresponding dynamics in region 2 still show diffusive characteristics and we therefore conclude that HDL transforms into LDL rather than LDA. The slowing down in region 3 can be explained by cooling after the heat release in combination with the difference in $T_g$ between the high- and low-density forms, respectively, 110 and 126 K. This observation is consistent with the dielectric relaxation measurements, where a deceleration by a factor of ~100 was observed upon transforming HDA to the low-density state (9). However, it cannot be excluded that the observed deceleration is also due to the formation of small cubic ice nuclei with a size much below 10 nm, thus preventing detection in WAXS (44). The alternative of revitrification into LDA form is not consistent with the observed diffusive dynamics (Fig. 4C) at the low-density form and therefore we derive that the final state corresponds to the LDL state. The diffusivity obtained here is the strongest indication of liquid-like motion between 115 and 130 K and evidence for HDL and LDL. During the HDL-to-LDL transition at 130 K, the diffusion coefficients slow down by nearly an order of magnitude, from 48 nm$^2$/s ($\sim 5 \times 10^{-17}$ m$^2$/s) for HDL to 5 nm$^2$/s for LDL ($\sim 5 \times 10^{-18}$ m$^2$/s). The latter is in fair agreement with previous experiments on ASW using CO$_2$ ($\sim 8 \times 10^{-17}$ m$^2$/s at 130 K) and molecular dynamics simulations ($\sim 1.5 \times 10^{-16}$ m$^2$/s at 135 K) (45). The diffusion of ASW at the liquid/ice interface was measured using nanoscale films ($\sim 10^{-20}$ m$^2$/s at 135 K) (21), which is considerably slower as the diffusivity in this case is presumably related to the mobility of molecules at the liquid/solid interface. Despite the difference in magnitude, the observed diffusivities exhibit a similar continuous trend across the ASW glass transition (21, 22). This interpretation was validated by measuring the LDA-to-LDL transition, where a diffusive component was observed at $T = 130$ K with nearly identical diffusion constants as those obtained in region 3 for LDL (SI Appendix, Fig. S9).

Summarizing our observations: (i) At temperatures $T < 110$ K a slow dynamical component is observed due to viscoelastic relaxation of HDA. (ii) Above 110 K a fast component appears exhibiting
diffusive character, which is attributed to the HDL. (iii) With increasing temperature HDA domains transform to HDL that further transform to the low-density form. (iv) During the high-to-low-density transition the dynamics slow down but remain diffusive, suggesting that the HDL is converting to LDL at 130 K. The question remains whether there exists also a stringent first-order phase transition at constant Gibbs free energy (8). Previous studies, varying pressure at temperatures above the glass transition, showed the presence of sharp boundaries between regions of HDA and LDA character (7, 46, 47). In accordance with these observations, our results point to the existence of a first-order phase transition between HDL and LDL, which can exist at higher temperature and under pressure (2). At ambient pressure, water’s structure varies continuously upon cooling (44), but if there is an HDL–LDL phase transition at higher pressure, then the structural evolution will be discontinuous when crossing the transition. This would indicate the existence of a thermodynamic point at elevated pressures where the two forms can no longer be distinguished (1).

Materials and Methods

Sample Environment and Preparation. HDA ice can be prepared through different routes within the metastable phase diagram, and several substrates that can be derived after annealing, heating, compression, and/or decompression at different temperatures have been named differently in the literature. We discriminate the preparation route of two most extreme cases, namely uHDA and eHDA ice. uHDA ice is prepared by direct pressure-induced amorphization of crystalline ice Ih at 77 K, i.e., without exceeding liquid nitrogen temperature (4, 6). The eHDA ice (7) is prepared by annealing uHDA ice at high pressure (1,100 MPa) with subsequent decompression to 70 MPa at 140 K, i.e., a region slightly above its glass-transition temperature Tg (9). This eHDA can be quench-recovered to ambient pressure and therefore studied as a metastable state within the stability region of LDA. With this experimental approach, we can probe the high-to-low-density transition by simply increasing the temperature at ambient pressure. The samples have been prepared by using a piston cylinder setup and a mechanical press (Zwick, Z100 TN).

For all measurements, a liquid N2 flow cryostat from JANIS was used. The amorphous ice samples were prepared and cold-loaded at liquid nitrogen temperature into the sample holder. A powdered sample was squeezed between two 50-μm-thick Kapton windows [WAXS at Advanced Photon Source (APS)] and 50-μm-thick diamond windows [XPCS at Deutsches Elektronen-Synchrotron (DESY)], respectively, resulting in a sample thickness of 2 mm (APS) or 1 mm (XPCS). The windows were held in place by a copper sample holder connected to the cold finger of the cryostat. The samples were measured in vacuum, i.e., without He-exchange gas. For the diffraction measurements at APS the cryostat was equipped with two Kapton windows and the pressure was P < 1 × 10−2 mbar. For the XPCS measurements the cryostat was connected directly to the beamline under vacuum without using additional windows.

WAXS Measurements and Data Analysis. X-ray diffraction measurements were performed at APS at beamline 6-ID-D using a 2D amorphous silicon area detector (Perkin-Elmer XRD1621) at Ε = 100 keV photon energy. CeO2 powder was used for sample–detector distance calibration (L = 348 mm) and Q calibration. The 2D diffraction image, showing the typical water ring, was angularly integrated by using the software Fit2D (V.17.006) after applying polarization and geometrical corrections. The scattering intensity I(Q) was corrected for background signal mainly from the Kapton windows and air measuring an empty sample holder and applied corrections for multiple scattering, detector efficiency, oblique incidence, and self-absorption (48). The static structure factor and the radial distribution function were calculated as described elsewhere (49). The I(Q) data were normalized to the molecular form factor and the Compton scattering was subtracted. The total structure factor S(Q) is given by

\[ S(Q) = 1 - \frac{I(Q) - FF(Q)}{WF(Q)} \]

where \( FF(Q) \) corresponds to the molecular form factor and \( WF(Q) \) to the weighting factor. The radial distribution function \( g(r) \) is calculated by a Fourier transformation of the static structure factor \( S(Q) \):

\[ g(r) = 1 + \frac{1}{2\pi^2} \int_0^Q dq \frac{M(Q, \Delta r)}{Q\Delta r} \frac{Q}{\sin(Qr)QdQ} \]

where \( \rho \) is the number density per Å³ and \( M(Q, \Delta r) \) is a Lorch-like function

\[ M(Q, \Delta r) = \frac{\sin(Qa)}{Q\Delta r} \]

where \( \Delta r \) is in the form

\[ \Delta r = a \left[ 1 - \exp \left( -\frac{r}{r_1} \right)^2 \right] + a_2 \left( 1 + \arctan \left( \frac{r_2}{r_2 + 2\pi} \right) \right)^{a_3}, \]

with parameters \( a = 2/\pi, r_1 = 2.8 \AA, r_2 = 0.5 \AA, a_2 = 12.0 \AA, \) and \( Q_{max} = 24.5 \AA^{-1} \).

XPCS Measurements and Data Analysis. XPCS measurements were performed at DESY, at the coherence application beamline P10 at the PETRA III synchrotron source, using a photon energy of 8.4 keV set by a channel-cut Si(333) monochromator. The WAXS patterns were recorded with a PILATUS 300K detector and the XPCS was performed with a Lambda detector. The following parameters were used: 5-m sample–distance (for SAXS), 3.4-μm horizontal and 2.8-μm vertical beam focus, exposure time 1 s, and 1,000 frames per measurement point. The corresponding photon counts for the lowest flux that was used can be assessed from SI Appendix, Fig. S5, with an average of ~0.5 photons per second. The reproducibility of the observed dynamics was verified by two independent beam times, where, during the first one, multiple sample positions were recorded for each temperature (10 sample positions per temperature separated by 100 μm) and thereby averaging in space, and during the second experiment the dynamics were recorded for longer times (up to 1,000 s) for each temperature.

The dynamics are calculated using the temporal intensity autocorrelation function:

\[ g_2(Q, t, \Delta t) = \frac{1}{N} \langle I(Q, t) I(Q, t + \Delta t) \rangle, \]

where \( \Delta t \) denotes the correlation time delay, \( I(Q, t) \) is the intensity at momentum transfer \( Q \), \( N \) the normalization factor \( N = \langle I(Q, t) \rangle^2 \), and the (…) indicates the average over all frames at different times \( t \). The 2D correlation maps were calculated by

\[ C(Q, t_1, t_2) = \frac{1}{N} \langle I(Q, t_1) I(Q, t_2) \rangle, \]

where the normalization factor is given by \( N = \langle I(Q, t_1) \rangle \langle I(Q, t_2) \rangle \). To estimate \( C \) in the frame \( \langle \Delta t, t \rangle \) we apply a 45° rotation to the \( (t_1, t_2) \) system transforming it to the \( \langle \Delta t, t \rangle \) reference frame and the corresponding maps are plotted logarithmically along \( \Delta t \) shown in Fig. 3C.

The functional form of the double-exponential function that was used to model the temporal autocorrelation is

\[ g_2(t) = A \cdot e^{-t/\tau_A} + (1 - A) \cdot e^{-t/\tau_B} + c, \]

where \( A \) is the amplitude bound to the interval \( [0, 1] \), \( \tau \) is the contrast parameter which for the given conditions is \( \beta = 0.4\pm0.02 \), and \( c \) is the offset ranging between \( [0.0, 0.02] \) which is attributed to residual scattering from streaks (SI Appendix). In addition, the flux dependence was investigated to ensure that the beam-induced heating is minimized (SI Appendix).

The Arrhenius analysis was performed by fitting the functional form \( f(t) = A \cdot e^{-E/RT} \), where \( R \) is the natural gas constant and \( E \) is the apparent activation energy; the fit results in activation barriers of 1.4 and 37.9 kJ/mol for the slow and fast component, respectively, where the component was fitted only for \( T \geq 110 \) K. The difference in activation barrier height reflects the different nature of the two processes as discussed here in the main text, with the latter being in fair agreement with the activation energy 34 kJ/mol obtained by dielectric measurements (9). The normalized variance \( \chi^2_T \) of the temporal autocorrelation function is a quantitative measure of the dynamical heterogeneities (38–40), which is an experimentally accessible estimator of the four-point dynamical susceptibility \( \chi_T \) (41). The variance is calculated along axis \( t \) by using the relation

\[ \chi^2_T(Q, \Delta t, t_\perp) = \left( \frac{C(Q, t_\perp, t_\perp)}{C(Q, t_\perp, t_\perp) + C(Q, t_\perp, t_\perp)} \right)^2, \]

where \( N \) is the normalization parameter \( N = \langle C(Q, t_\perp, t_\perp) \rangle \). The normalized

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variance is shown in Fig. 4B, where the peak position follows the trend exhibited by the result of the exponential fit (the position should be se for a single-exponential fit) and the peak amplitude quantifies the heterogeneities. At \( T = 89 \) K, we observed a double peak, which can be related to the underlying multiple dynamical components, assigned as fast and slow dynamics in the main text. Interestingly those are not evident in the \( g_2 \) but can be seen as separate peaks in the normalized variance \( \chi^2 \).

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